New Gel Polyelectrolytes for Rechargeable Lithium Batteries

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Abstract: New polyelectrolytes were synthesized by grafting the allyl group containing lithium

salt, lithium bis(allylmalonato) borate (LiBAMB), onto poly[poly(ethylene glycol) methyl ether

acrylate-co-allyoxyethyl acrylatel through hydrosilylation chemistry. Gel polyelectrolytes were

obtained by adding 50wt% of different 1/1 (wt/wt) mixtures of propylene carbonate (PC),

ethylene carbonate (EC), dimethyl carbonate (DMC) and tetraethyleneglycol dimethyl ether

(TEGDME). The highest ambient conductivity was 2.7 x 10⁻⁸ S.cm⁻¹ for the dry single ion

conductors and 7.9 x 10⁻⁶ S.cm⁻¹ for the gel single ion conductor containing 50 wt% of EC/DMC

(1/1, wt/wt) and both are obtained for the sample with an EO/Li ratio of 40:1. The conductivity

order of gel electrolytes containing the same amount of different mixed solvents suggests that the

conductivity of the gel is more determined by the dielectric constant rather than by the viscosity

of the solvent. The preliminary Li/Li cycling profile of a dry single ion conductor is encouraging,

as almost no concentration polarization and relaxation was observed. However, some fluctuation

of potential occurred, which might be due to the reactions of the electrolyte on the surface of

lithium metal.

Keywords: Single ion conductors, ionic conductivity, plasticizer, gel electrolytes, cell cycling

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1. Introduction

System modeling results have shown that in applications such as electric vehicles, where a high discharge rate is desired, electrolytes with a Li⁺ ion transference number equal to unity can substantially improve active material utilization [1]. As a result interest is growing in the synthesis and characterization of new single ion conductors. Most single ion conductors are synthesized by fixing either alkyl sulfonate [2] or carboxylate [3] to the polymer backbones. However, due to the limited solubility and ion dissociation in polyether media, the ambient conductivities of these materials are usually in the range of 10⁻⁷-10⁻⁸ S.cm⁻¹. In order to enhance the ionic dissociation, different anions have been synthesized [4, 5] and ambient conductivities of 10⁻⁶-10⁻⁷ S.cm⁻¹ have been realized. Shriver and coworkers [6] reported ambient conductivity of 10⁻⁵ S.cm⁻¹ for lithium conductive poly (aluminosilicate). In addition, plasticizers have enhanced the ambient conductivity of single ion conductors to the range of 10⁻⁴ S.cm⁻¹[7].

In this paper we report the synthesis of a new kind of single ion conductor based on a new lithium salt, lithium bis(allylmalonato)borate (LiBAMB), and an allyl group containing comb branch polyacrylate ether (Scheme 1). Since LiBAMB has allyl groups on both sides, after connection to the polymer chain through tetramethyldisiloxane by hydrosilylation it not only acts as a lithium source but also acts as a cross-linking site to enhance the mechanical properties of the resulting new single ion conductors. Plasticizers such as mixtures of PC, EC, DMC and TEGDME were used to obtain the corresponding gel single ion conductors.

2. Experimental

The synthesis of LiBAMB, shown in Scheme 1, follows the literature procedure [8, 9]. The structures of the synthesized monomers, prepolymers as well as the final single ion conductors were confirmed by NMR and full details will be given elsewhere [10]. Glass

transition temperatures (T_g) were measured using a Perkin Elmer-7 DSC instrument. The heating rate was kept at 10° C/min. The T_g is defined as the heat capacity jump onset temperature. The melting temperature T_m was obtained from the peak temperature of the melting process. The measurement of transport properties was carried out according to the method of Newman and coworkers [11,12]. Conductivity measurements AC impedance were made with blocking stainless steel electrodes in Swagelok cells Li/Li cells were constructed inside a Helium filled dry box and the cells were cycled at 85°C at different current densities.

3. Results and Discussion

The general physical properties of the prepolymers, the dry single ion conductors as well as those of the gel single ion conductors are summarized in Table 1. For prepolymers with increasing the content of crosslinker (lower x value), 2-allyoxyethyl acrylate, the molecular weight increases and the corresponding glass transition temperature also increases. After crosslinking by LiBAMB through hydrosilylation, the glass transition temperatures of the resulting dry single ion conductors are all higher than those of the corresponding prepolymers. Here the increase of glass transition temperature is not only due to the crosslinking by the bridging lithium salt but also due to the complexation of lithium cation by carboxylate group in the anion and oligoether side chains. As expected after plasticizing with 50wt% EC/DMC (1/1,wt/wt), all the glass transition temperatures of the gel single ion conductors are lower than those of the corresponding dry single ion conductors by at least 20°C.

The ionic conductivities of dry single ion conductors shown in Fig. 1 are disappointedly low, which are mainly due to the severe ion pairing effect between the lithium ion and the BMB anion. The highest ambient conductivity of $2.7 \times 10^{-8} \text{ S.cm}^{-1}$ is obtained for the sample with an EO/Li of 40/1. The maximum in conductivity is due to the trade off in the effects of chain

flexibility versus charge carrier concentration. Higher charge carrier concentration results in more lithium ions moving but also restricts the motion of the polymer chains due to ionic cross-linking which results in an increase in the T_g and lower mobility. This effect apparently extends to the gel systems also. The addition of 50-wt% EC/DMC (1/1,wt/wt) to the dry single ion conductors not only increases mobility of the salt but also is likely to dissociate more ion pairs into free ions. These combined effects result in an increase of more than two orders of magnitude higher conductivity of the resulting gel single ion conductor over the corresponding dry single ion conductor. Raman and FT-IR spectroscopy is presently being applied to investigate the extent of ion dissociation. The highest ambient conductivity of 7.9 x 10^{-6} S.cm⁻¹ for gel single ion conductor is obtained for the same sample with EO/Li ratio of 40/1.

Fig. 2 shows the solvent effect on ionic conductivities of gel single ion conductors that were formed from the same dry sample, EO/Li=10, and 50 wt% of a variety of solvents mixed in 1/1 (wt/wt) ratio. For the mixture of PC/EC that has the highest dielectric constant and highest viscosity, the corresponding gel shows the highest conductivity. For the mixture of DMC/TEGDME that has the lowest dielectric constant and medium viscosity, the corresponding gel shows the lowest conductivity. In the middle, the mixture of EC/DMC has a similar viscosity to that of PC/DMC and the mixture of EC/TEGDME has similar viscosity as that of PC/TEGDME. However, the gels containing the EC mixture show higher conductivity than those containing the PC mixture. In addition, the TEGDME containing mixtures such as EC/TEGDME and PC/TEGDME have higher viscosity and higher dielectric constant than DMC containing mixtures such as EC/DMC and PC/DMC but the gels containing TEGDME mixture show higher conductivity than those gels containing DMC mixture. These observations seem to

induicate that for gel single ion conductors the conductivity is more determined by the dielectric

constant rather than by the viscosity of the solvent.

The dry single ion conductor with a salt concentration of EO/Li=80/1 was used to

construct a Li/Li cell, which was cycled under the sequence of 1hr relaxation, 2hr charge, 1hr

relaxation, 2hr discharge at a current density of 50µA.cm⁻². The preliminary result shown in

Fig.3 is encouraging, since as expected for single ion conductors, almost no concentration

polarization and relaxation was observed. It can be observed that the application of the current

results in an immediate polarization of the voltage which then decreases with further current.

Concentration polarization would result in a further increase in voltage after the immediate

instantaneous polarization. The relatively high initial potential is caused by large bulk and

interfacial impedances (not shown here). The gradual increase of potential with cycling is due to

the increase of interfacial impedance, which is apparently related to the buildup of the SEI layer.

The potential fluctuation within one cycle might be related to some undesired reactions on the

surface of lithium metal, since trace amount of one end free silane from uncompleted

crosslinking steps might still present in the electrolytes and the hydrogen in the methide group

between the two carbonyls of the salt is also more active. These are all susceptible to attack by

lithium metal at a temperature as high as 85°C. Further studies are underway to elucidate the

mechanism in more detail.

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Scheme1: Synthesis of network type single ion conductors

Table 1. Summary of physical properties of prepolymers, dry single ion conductors and gel single ion conductors.

No		I	II	III	IV
	X	0.8	1.8	3.8	7.8
	Mw (x 10 ⁴)	11.7	3.12	2.20	1.48
Prepolymers	PDI	1.78	2.31	1.88	1.59
	T _g /°C	-44.8	-59.5	-63.0	-64.7
Dry single	EO/Li	10/1	20/1	40/1	80/1
Ion Conductors	T _g /°C	-37.2	-52.5	-54.7	-55.4
	$Log\sigma_{25}$		-7.94	-7.57	-7.96
Gel Single Ion Conductors	EO/Li	10/1	20/1	40/1	80/1
with 50 wt%	T _m /°C	35.1	24.1	-21.8; 5.9	-18.8; 17.1
EC/DMC (1/1, by wt)	T _g /°C	-58.2	-74.0	-80.4	-86.1
	$Log\sigma_{25}$	-5.97	-5.27	-5.10	-5.51

Note: DSC scan of EC/DMC(1/1, wt/wt) shows two large peaks at -16 and 16° C and the two small melting peaks of pure DMC and EC at around 0° C and 36° C respectively.

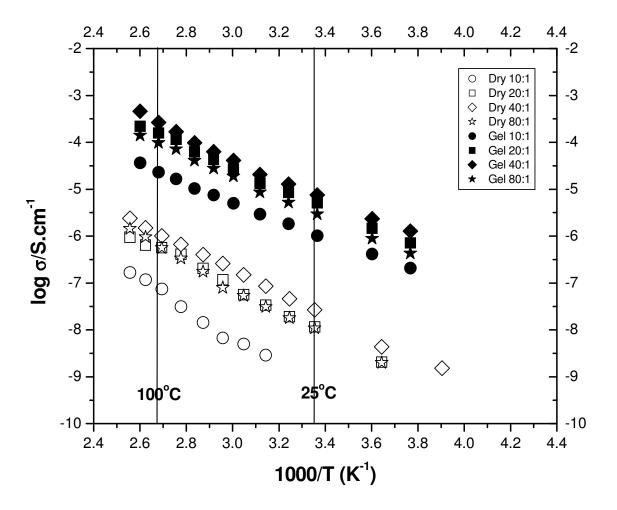


Fig. 1. Arrhenius plots of ionic conductivities of dry single ion conductors and those of gel single ion conductors containing 50 wt% of EC/DMC (1/1, wt/wt).

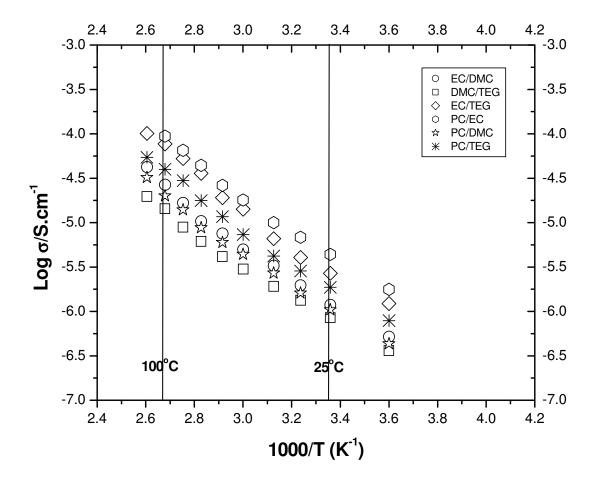
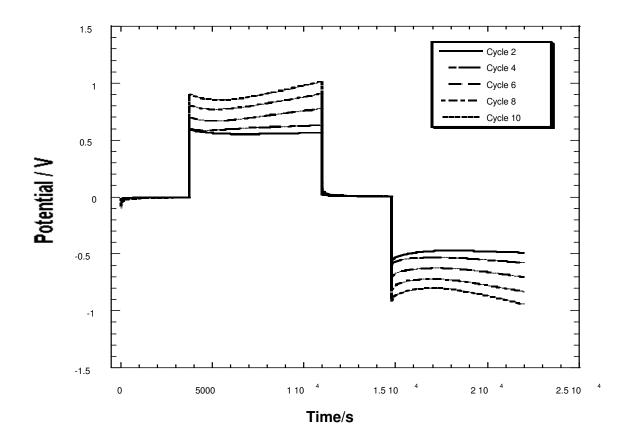


Fig. 2. Arrhenius plots of ionic conductivities of the dry single ion conductor with salt content of EO/Li=10 containing 50 wt% of different mixed plasticizers (1/1, wt/wt).



Φιγ. 3. Χψχλινη προφιλε της Λι/Λι χελλ οφ δρψ σινήλε ιον χονδυχτορ ωιτη α σαλτ χονχε ντρατιον οφ ΕΟ/Λι=80/1 υνδερ της σεθυενχε οφ 1ηρ ρελαξατιον, 2ηρ χηαρής, 1ηρ ρελαξ ατιον, 2ηρ δισχηαρής ατ α χυρρεντ δενσιτψ οφ $50\mu\text{A.cm}^{-2}$.